

# Enhancement of Imprinting Effect in Ferroelectric AlScN Films for Non-Destructive Capacitive Readout Operation Using Metal-Capping Oxidation (MCO) Process

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## 1. Introduction

The concept of capacitive readout for ferroelectric (FE) memory technology has recently drawn considerable attention for non-volatile memory (NVM) [1]. Unlike the traditional switching-driven readout scheme [2], the capacitive readout features lower power consumption and non-destructive data sensing, which alleviates the burden on endurance strength. However, the key to realizing capacitive readout is that the memory window (MW), i.e., high/low capacitance gap at 0 V, should be sufficiently wide. Although some experimental demonstrations have been conducted with HfO<sub>2</sub>-based FE materials [3-4], research works with capacitive readout are still lacking for nitride FE materials such as AlScN. With a lower relative permittivity ( $\epsilon_r$ ) of 11-19, AlScN holds a great advantage over HfO<sub>2</sub>-based and lead zirconium titanate (PZT) FE materials with this scheme. Herein, we propose a simple metal-capping oxidation (MCO) process for AlScN thin films with enhanced MW, thereby unlocking their potential for capacitive readout operation.

## 2. Experimental Methods

The structural diagram and process flow of AlScN capacitors are shown in Figs. 1(a) and 1(b), respectively. The experimental procedures are similar to our previous work [5], except that we fabricated two FE layer thicknesses (25 and 50 nm). Additionally, we deployed the MCO process before top electrode deposition to indirectly oxidize the FE layer [6], as described below. After FE deposition, a 3-nm W layer was capped on the FE layer and followed by rapid thermal annealing (RTA) in O<sub>2</sub> ambient for 30 min at 400°C, which transforms metal W to dielectric WO<sub>x</sub>.

## 3. Results and Discussion

Fig. 2 depicts the polarization-electric field (P-E) hysteresis and current density-electric field (J-E) curves for all capacitors, validating the FE characteristics. However, due to leakage current, positive-up negative-down (PUND) was deployed, as displayed in Figs. 3(a) and 3(b). For 50-nm FE films, P<sub>r</sub> is enhanced at higher fields and exhibits good semi-saturation by the MCO process. Similarly, for 25-nm FE films, considerably enhanced P<sub>r</sub> shows that the MCO process could compensate for the deficit of worse crystallinity in thinner films [7], and thereby reaching 191  $\mu\text{C}/\text{cm}^2$  in maximum. Moreover, a slight increase in E<sub>c</sub> values is observed in MCO-treated capacitors, as plotted in Fig. 3(c). Fig. 4 plots the leakage and breakdown characteristics for all capacitors in the as-deposited state. Leakage suppression and breakdown field (E<sub>BD</sub>) improvements are observed in MCO cases, though the source of contribution still remains to be distinguished. Involvement of the dielectric WO<sub>x</sub> layer or partially oxidized AlScN FE layer could be the origin. Additionally, the switching event becomes more pronounced for MCO capacitors, indicating that the initial N-polar state can be partially inverted by extrinsic oxidation, in addition to incorporation of oxygen atoms during AlScN deposition [8].

Fig. 5 displays the butterfly-shaped capacitance-electric field (C-E) curves. Overall,  $\epsilon_r$  is generally decreased with the MCO process, which is attributed to the insertion of a thin dielectric WO<sub>x</sub> layer. The permittivity at the cross-point ( $\epsilon_i$ ) is extracted and plotted in Fig. 6(a). The imprint effect is visible with MCO process, as  $\epsilon_i$  values shift toward negative, and possibly due to high defect density at the oxide/nitride interface. For capacitive readout, the MW is defined here as  $\Delta C$  and  $\Delta C^*$ , respectively, for as-deposited and 100-cycle-switched capacitors, as indicated in Fig. 5(d). With the MCO process,  $\Delta C$  is significantly increased for both 25 nm and 50 nm capacitors. Furthermore,  $\Delta C^*$  also demonstrated robustness against write operations. It is speculated that the uprising  $\Delta C$  and  $\Delta C^*$  with MCO process are related to a modified switching mechanism by graded oxidization of FE AlScN. Nonetheless,  $\sim 4.2\times$  enhancement of  $\Delta C$  and  $\Delta C^*$  is demonstrated, as summarized in Fig. 6(b).

## 4. Conclusions

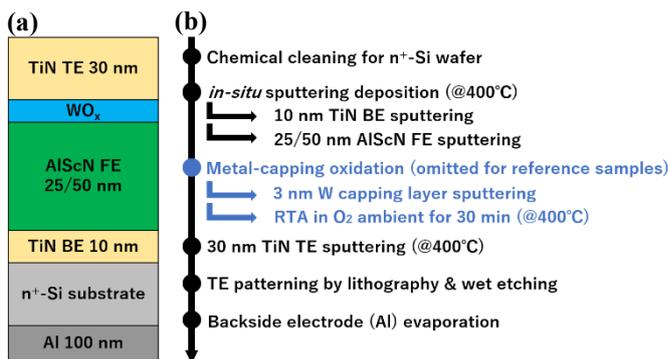
In this work, we study the utilization of the MCO process in FE AlScN thin films to unlock the potential of non-destructive capacitive readout in nitride FE materials. The results indicate that MCO not only boosts P<sub>r</sub> performance but also suppresses leakage current while enhancing E<sub>BD</sub>. Owing to the thin dielectric WO<sub>x</sub> layer,  $\epsilon_r$  is found to be slightly decreased along with more pronounced imprint effects. Most importantly, MCO-treated FE AlScN films exhibit  $\sim 4.2\times$  increment of MW at zero electric field, which is beneficial for capacitive readout operation in future NVM technologies.

## Acknowledgements

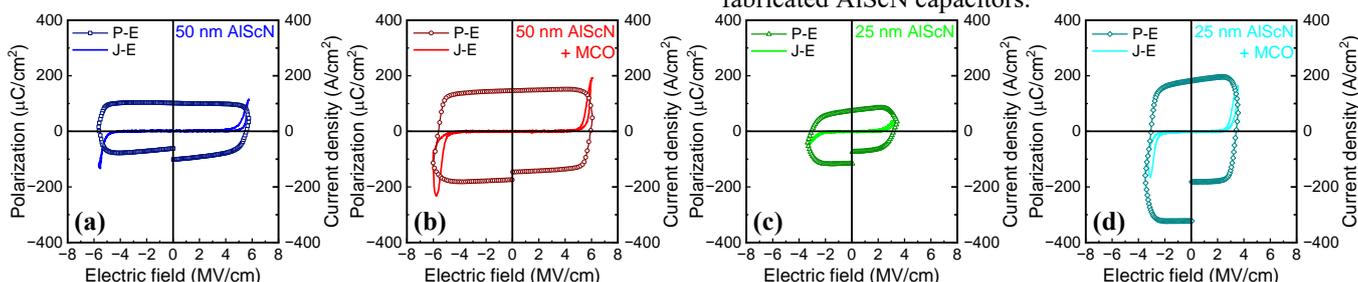
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## References

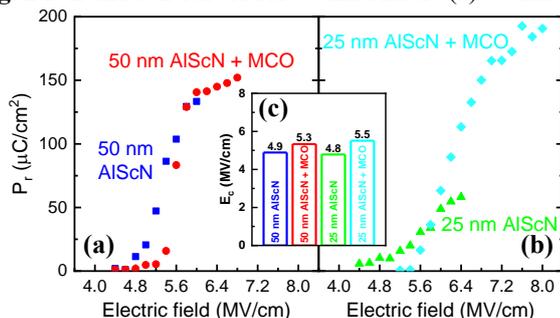
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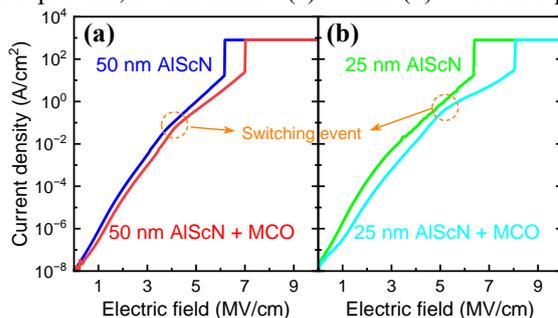
**Fig. 1** (a) Schematic diagram and (b) process flow of fabricated AlScN capacitors.



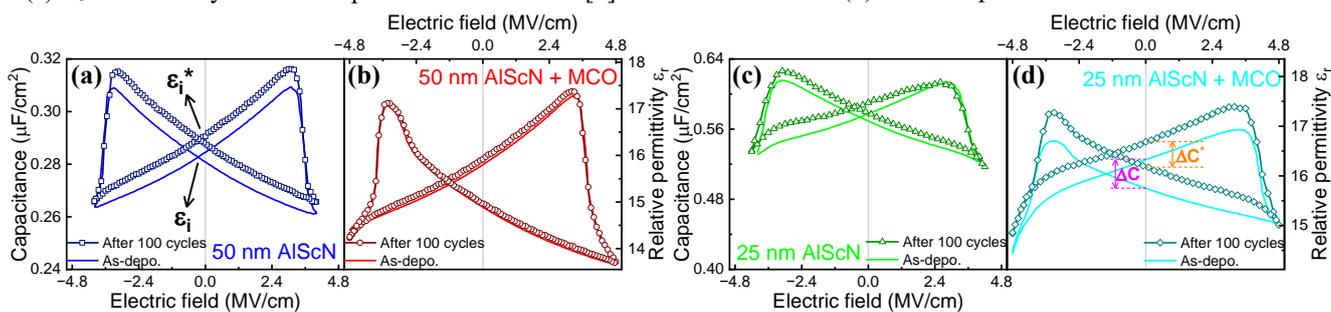
**Fig. 2** P-E and J-E curves for 50 nm AlScN (a) w/ and (b) w/o MCO process; 25 nm AlScN (c) w/ and (d) w/o MCO process.



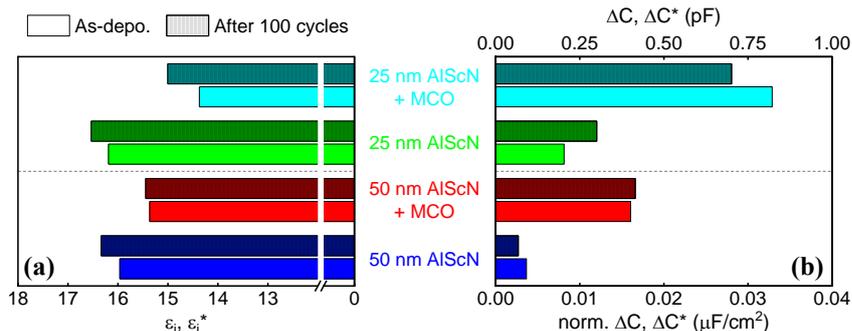
**Fig. 3** PUND results for (a) 50 nm and (b) 25 nm capacitors. (c)  $E_c$  extracted by linear extrapolation from PUND [5].



**Fig. 4** Leakage and breakdown characteristics for (a) 50 nm and (b) 25 nm capacitors.



**Fig. 5** Butterfly-shaped C-E curves for 50 nm AlScN (a) w/ and (b) w/o MCO process; 25 nm AlScN (c) w/ and (d) w/o MCO process. Solid lines and scattered points represent data from as-deposited and 100-cycle-switched capacitors.



**Fig. 6** Comparison of (a)  $\epsilon_i$  and (b)  $\Delta C$  among all capacitors.